

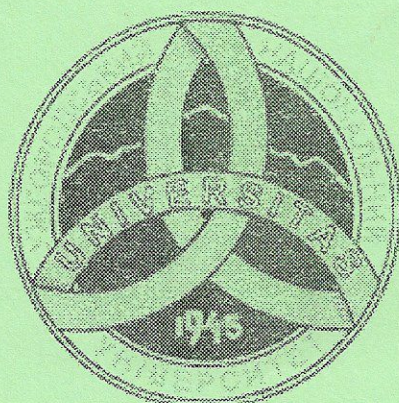
**6<sup>th</sup> SEMINAR**  
**Properties of ferroelectric and  
superionic systems**

Program and abstracts

**6-й СЕМІНАР**  
**Властивості сегнетоелектричних  
і суперіонних систем**

Програма і тези доповідей

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Ужгород – Україна  
Uzhhorod – Ukraine

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# STRUCTURAL DISORDER AND OPTICAL PROPERTIES OF ELECTRON-IRRADIATED $As_2S_3(Se_3)$ CHALCOGENIDE GLASSES

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Glassy chalcogenides of arsenium are characterized by high transparency in a near and middle infrared and belong to a class of materials which are used as active or passive elements in optical engineering]. Experimental studies of the influence of radiation load (gamma radiation, X-radiation or electron radiation) enable to determine the character and change of physical properties of these materials, boundary doses of radiation, to learn the nature and mechanisms of radiational defect – formation, reveal conditions of renewing the initial parameters.

The samples radiation was carried out in usual conditions at room temperature with electrons of an average energy  $\sim 0.6$  MeV; the radiation source was SrY-16 of 12,2 Curi activity and the flux density  $10^7$  electrons/( $cm^2 \cdot s$ ). The absorbed dose was calculated starting with the expositional dose and the radiation time in a definite point of the source channel. The absorbed dose value was choosed according to a well-known results of dose dependences of radiationally-induced effects in glassy chalcogenide glasses of  $As_2S_3$  type.

Gody et al. [1] were the first to determine experimentally the proportionality  $E_g^*(T, X)$  and  $W(T, X)$  for a- Si:H, which was realized in the whole range of  $W$  values studied till now. By using the Tauc's concept of "freezed" phonons he spread the idea of the equivalency of the effect of a structural  $W_s$  and thermal  $W_t$  of disorder onto the band width  $E_g^*$  and got a linear relation between  $E_g^*$  and  $W$ :

$$E_g^*(T, X) = E_g^*(0,0) + D \cdot \langle W^2 \rangle_0 - \frac{D}{K} W(T, X),$$

where  $D$  is a deformation potential,  $\langle W^2 \rangle_0$  – mean-square shift due to zero oscillations. According to this model the optical pseudogap  $E_g^*(T, X)$  is determined by the degree of disordering of a glass lattice which is described by  $\langle W^2 \rangle_s$  parameter, ie. By changing it by sources of different nature it is possible to influence the  $E_g^*$  value indirectly.

Let us analyze our experimental results in the frameworks of this model. In Fig.1 the correlation between  $E_g^*$  and  $W$  for glassy  $As_2S_3(Se_3)$  in dependence of the nature of disorder due

to various external factors. This correlation shows that the optical pseudogap  $E_g^*$  and – this being more important – the slope of an exponential portion of the edge are changing in dependence of the disorder degree. A linear relation between  $E_g^*$  and  $W$  for chalcogenide glasses  $As_2S_3(Se_3)$  is fulfilled practically in the whole range of the values of  $W$  energies which was studied up to this time. Thus it can be stated that in this case for these materials the contribution of the structural (“intrinsic” and induced) and the thermal contributions into a change of disorder potential is adequate, and the change of the slope probably reflects the change of the distribution of the states in the tails of zones.

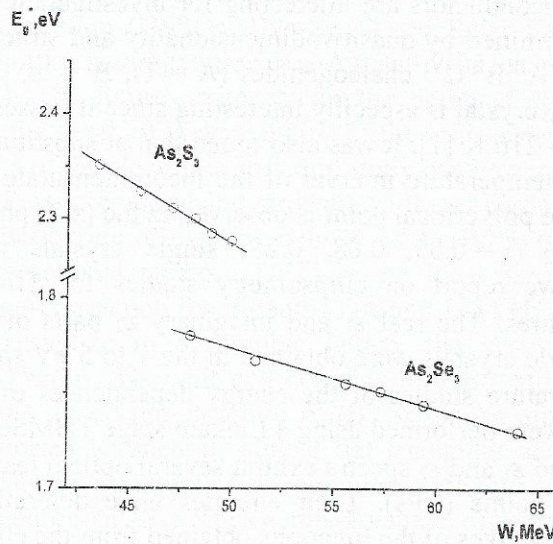


Fig.1. The dependence of optical pseudogap  $E_g^*$  on the slope of the exponential Urbach tail  $W$ .

Dose dependences of energy parameters of the intrinsic absorption edge testify to an electron-induced creation of new defects which change the disorder potential. A characteristic energy of the exponential absorption tail  $W(T, X)$  shows not only the temperature but also the structural disordering of other kinds:

- a) intrinsic structural disorder of an “ideal” glass;
- b) induced structural disorder due to external factors (of radiation or technological nature).

1. Gody G.D., Tiedje T., Abeles B., Brooks B., Goldstain Y. Phys. Rev. Letters 47 (1981) 1480.